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### International Journal of Polymeric Materials

Publication details, including instructions for authors and subscription information: <http://www.informaworld.com/smpp/title~content=t713647664>

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To cite this Article Privalko, V. P. , Buhrmistr, M. V. and Lipatov, Yu. S.(1993) 'Thermodynamic Properties and Flow Behavior of Polyionene Solutions', International Journal of Polymeric Materials, 21: 1, 93 — 102 To link to this Article: DOI: 10.1080/00914039308048517 URL: <http://dx.doi.org/10.1080/00914039308048517>

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# Thermodynamic Properties and Flow Behavior of Polyionene Solutions

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**The effect of temperature, chain length and concentration on the conformational state and intermolecular interactions of two model polyionenes in water solutions is discussed.** 

**KEY WORDS Polyionene solutions, rheology. conformational states.** 

#### **INTRODUCTION**

Polyionene (PI) is a generic name for a special class of polyelectrolytes with positive charges in the chain backbone and either equal (symmetrical PI) or unequal (unsymmetrical PI) spacings between charges.<sup>1</sup> In recent years a renewed interest in PI's was stimulated by novel data on their behavior both in the bulk state $^{2,3}$  as well as in solution.<sup>4-6</sup> For example, as can be inferred from the viscosity studies of PI solutions in water and in mixed solvents,<sup>5,6</sup> the structure of the solvent is changed on addition of even very small amounts of PI. It is thus the purpose of the present paper to elucidate the effect of temperature, chain length and concentration on the conformational state and intermolecular interactions in water solutions of two model PI's.

#### **EXPERIMENTAL**

#### **Materials**

Two model unsymmetrical PI's of the general structure



were synthesized by methods described in detail elsewhere.' Polymers were recovered from reaction medium by precipitation with acetone and dried in vacuum at 323 K for several hours to constant mass. Z-Average molecular masses  $\langle M_2 \rangle$ , determined from sedimentation data by ultracentrifuge model **MOM-3180.** were  $17.9 \times 10^3$  (PI-1) and 8.4  $\times 10^3$  (PI-2).

#### **Methods**

Integral heats of solution  $(\Delta H_s)$  and dilution  $(\Delta H_d)$  were measured (with mean error **1.0-** 1.5%) with the aid of a home-made differential calorimeter on diathermal cells with an in-built microstirrer and calibrated electric heater.<sup>8</sup> Most of the data presented in the subsequent discussion are the mean values taken from two or three independent measurements.

Viscosities of dilute solutions were measured (with mean error 0.5- **1** .O%) with an Ubbelohde-type viscometer  $(t = 150 \text{ sec for water at } 298 \text{ K})$ ; at higher concentrations a constant shear rate-type rotational instrument, model RHEOTEST-**2.1,** with a double-cylinder geometry, was used (mean error on the order of 5%).

#### **RESULTS AND DISCUSSION**

#### **Thermodynamic** Properties

**As** can be seen from Figure 1, all isothermal plots of concentration dependence of specific (i.e., per unit mass of a polymer) heats of solution,  $\Delta H_{\rm g}$ , exhibit minima, the positions of which are functions of temperature. The patterns of temperature dependence of  $\Delta H_s$  at constant mass fraction of a polymer,  $W_2$ , are drastically different in temperature intervals below and above **303** K, namely (see Figure 2): at  $T < 303$  K the *exothermic* ( $\Delta H_s < 0$ ) heat effects increase linearly with temperature, whereas at  $T > 303$  K a sudden transition to *endothermic* ( $\Delta H_s > 0$ ) effects is observed. The dilution phenomena at different initial concentrations of PI in solution (Figure 3) are also endothermic  $(\Delta H_d > 0)$ .



FIGURE 1 Concentration dependence of heats of solution.



**FIGURE 2** Concentration dependence of heats of dilution at  $T = 295$  K.



**FIGURE 3** Temperature dependence of heats of solution at  $W_2 = 2.3 \cdot 10$ .

![](_page_3_Picture_140.jpeg)

**FIGURE 4** Scheme **1:** Contributions to the heat **of** interaction **of** polyionenes with water.

The above data will be analyzed within the framework of the following structural considerations. Macromolecules of bulk PI obtained by precipitation from a good solvent (reaction medium), are in a coiled conformation which satisfies the conditions of both screening of intramolecular repulsion between charged atoms and saturation of all possible kinds of intermolecular bonds. Obviously, as more and more solvent is added to polymer, both local and overall concentrations of counterions will be continuously decreasing ("intermolecular dilution"). while local (intramolecular) concentration of positive charges in the chain will remain unchanged. Thus, in the "infinitely dilute" solution regime the effect of intramolecular screening vanishes and the macromolecules undergo a transition into extended

![](_page_4_Figure_1.jpeg)

FIGURE 5 Concentration dependence of reduced viscosity at  $T = 295$  K.

![](_page_4_Figure_3.jpeg)

**FIGURE 6 Temperature dependence of shear viscosity at indicated polymer concentrations.** 

![](_page_5_Figure_1.jpeg)

**FIGURE 7 Dependence of shear viscosity on shear stress at indicated temperatures and concentrations.** 

conformation (effect of polyelectrolyte swelling) which corresponds to a maximum possible distance between charged atoms.9 Transition to a "semi-dilute" solution regime would lead to the onset of electrostatic interactions between extended chains whereas further increase of polymer concentration (i.e., transition to a "concentrated" solution regime) would strengthen the effect of interchain screening of electrostatic interactions and, consequently, induce the transition of macromolecules into more coiled conformation.<sup>10,11</sup>

Using Scheme 1 (Figure **4)** as a starting point for a qualitative discussion of calorimetric data (Figures **1-3)** it may be argued that the exothermic pattern of dissolution of both **PI'S** below **303** K (Figures **1** .and **2)** is the manifestation of a dominant contribution of exothermic effects  $\Delta H^E$ ,  $\Delta H^{H+}$ ,  $\Delta H^{H-}$  as compared to  $\Delta H^V$  and  $\Delta H^D$ , which accompanied the transition of PI chains from the bulk state into dilute solution. The more exothermic pattern of dissolution at increased dilution (Figure 1) may be explained by a smaller contribution of  $\Delta H^D$ , whereas subsequent (at  $W_2 < 4.10^{-3}$ ) decrease of "exothermicity" may be explained by a lower degree of intermolecular association of extended PI chains (and, correspond-

![](_page_6_Figure_1.jpeg)

**FIGURE 8** Concentration dependence of shear viscosity at  $T = 298$  K and constant molecular mass.

ingly, by a smaller contribution of  $\Delta H^{H+}$ ,  $\Delta H^{H-}$ ) at a negligibly small contribution from the effect of breakdown of the structure of bulk water *(AHD).* 

Negligible values of the heats of dilution  $\Delta H_d$  in the range of fairly small concentrations  $(W_2 < 0.05)$  of both PI's in initial solutions (Figure 3) are also an indication of a negligible contribution of  $\Delta H^D$  to interactions of water molecules with individual macromolecules of PI in an extended conformation. Onset of the endothermic heats of dilution at higher initial concentrations  $(W_2 > 0.05)$  is likely a manifestation of a transition to the semi-dilute solution regime with increased contribution from  $\Delta H^D$ , while subsequent accelerated rise of  $\Delta H_d$  at  $W_2 > 0.20$  is presumably due to a transition to the concentrated solution regime where a continuous gain of endothermic  $\Delta H^D$  and  $\Delta H^V$  effects exceeds by far the contributions of exothermic  $\Delta H^E$ ,  $\Delta H^{H+}$  and  $\Delta H^{H-}$  processes.

In conclusion, a discussion of the temperature effect on the energetics of PI dissolution (Figure 2) is in order. Assuming that the final concentration,  $W_2 =$  $2.3 \cdot 10^{-3}$ , corresponds to the dilute solution regime (see above), it may be argued that the observed sudden change of the pattern of temperature dependence of  $\Delta H$ , at T = **303 K** is **a** result of a structural transformation in either a solute phase (PI)

![](_page_7_Figure_1.jpeg)

**FIGURE 9 Molecular mass dependence of shear viscosity of PI-l solutions at** *T* = **298 K and constant concentration.** 

or a solvent phase (water). **As** follows from Scheme **1,** at low polymer concentration the major contributions to  $\Delta H$ , are provided by two exothermic hydration processes, *AHH+* and *AHH-,* and by an endothermic process of water structure breakdown,  $\Delta H^D$ . Therefore, positive slopes on the  $\Delta H_s$  versus T plots (in other words, positive values of the apparent heat capacity of solution process,  $C_{p,s} = d\Delta H_s/dT$ ) above  $T = 303$  K may be interpreted as qualitative evidence of coiled conformation of PI chains, an arrangement which discourages interactions of both hydrophilic and hydrophobic groups of PI with water molecules (hence, the contributions of *AHH+*  and  $\Delta H^{H-}$  are small) but induces the change of water structure (the contribution of  $\Delta H^D$  dominates). On the other hand, a negative value of  $C_{p,s}$  at  $T < 303$  K indicates easier access of water molecules to hydrophilic and hydrophobic sites in PI chains as a result of transition of the latter into more extended conformation (hence, contributions of  $\Delta H^{H+}$  and  $\delta H^{H-}$  increase) which perturbs the water structure less (contribution of  $\Delta H^D$  decreases).

**A** similar conclusion may be reached starting from completely different assumptions. It can be easily shown that the contour distance *A* between charges in **PI** chains  $(A = 0.7 \text{ nm}$  and  $A = 0.6 \text{ nm}$  for PI-1 and PI-2, respectively) are of the same order of magnitude as the characteristic Bjerrum length  $l_B^9$  for the formation of ionic pairs in water at  $T = 303$  K ( $l_B = e^2/DkT = 0.7$  nm), but is smaller than the effective thickness of the screening ionic atmosphere<sup>11</sup>:  $(k'')^{-1} = (4\pi A W_2)^{-2}$  $= 6.5$  nm for PI-1 and 6.0 nm for PI-2 where  $e^2 = 2.3 \cdot 10^{-26}$  J $\cdot$ cm is the square of electron charge,  $D = 78.5$  is the dielectric permittivity of water,<sup>12.13</sup> and *k* is Boltzmann's constant). It follows that the electrostatic chain persistence length,'  $L^E = l_R/4(k'')^2A^2$ , with values of 15 nm for PI-1 and 18 nm for PI-2, is larger than the thickness of the screening ionic layer at  $T = 303$  K. Taking into consideration that cooling to  $T = 278$  K is accompanied by a nearly 1.5-fold drop of dielectric permittivity of water,<sup>14</sup> it can be shown that this effect should lead to a corresponding increase of  $l_B$  up to 1 nm, and an increase of electrostatic chain persistence length up to **21.5** nm for **PI-1** and **25.7** nm for **PI-2.** 

Judging by the above estimates, in the temperature interval above **303 K** the macromolecules of **PI's** in dilute water solutions are in a coiled state which is favorable for the endothermic pattern of breakdown of the structure of bulk water. Cooling of such dilute solutions below **303 K** results in the transition of **PI** chains into a more extended conformation which strengthens the contributions from exothermic processes of hydrophilic and hydrophobic hydration and minimizes the effect of breakdown of the structure of bulk water.

#### **Flow Behavior**

**As** can be seen from Figure 5 (left-hand side), both **PI's** exhibit a classical polyelectrolyte behavior, $4.5.9$  namely, at extremely small polymer concentrations further dilution leads first to the drastic increase of reduced viscosity,  $\eta_{\rm s}/C$ , followed by a substantial drop of this viscosity below the first "critical" concentration,  $W_2^{1u}$  =  $5 \cdot 10^{-5}$ , which marks the transition from the "dilute" to "extremely dilute" solution regime. The former phenomenon may be explained by chain extension as a result of reduced shielding of positive charges in the chain backbone by the counterions (effect of intramolecular repulsion between identical charges), while the latter one simply reflects the reduced interactions between extended macromolecules (intermolecular effect).

**AS** the concentration grows (Figure 5, right-hand side), the reduced viscosity slightly decreases (presumably because macromolecules gradually become less extended) up to the second "critical" concentration,  $W_2^2 =$  (with values of 0.05 and **0.08** for **PI-1** and **PI-2,** respectively), and then starts to increase again (due to the onset of aggregation of less extended **PI** chains). It is pertinent to remark here that it is only below  $W_2^2$ <sup>u</sup> that one observes the breaks at about  $T = 303$  K on the semilogarithmic plots of shear viscosity of **PI-1** solutions against reciprocal temperature (Figure 6). Recalling the data on the temperature dependence of heats of solution (Figure 4) it may be argued that at  $W_2 < W_2^{2\mu}$  (i.e., in the "dilute" solution regime) the lower activation energies for viscous flow above  $T = 303$  K ( $\Delta E_V/k = 1.3 \times$ **103.K)** refer to the flow of coiled, individual **PI-1** chains, whereas more than a two-

fold increase of  $\Delta E_v/k$  below  $T = 303$  K (up to 3.1  $\times$  10<sup>3</sup> K) reflects the transition of PI-1 macromolecules into a more extended conformation.

On further increase of  $W_2$ , the breaks on the above plot become less pronounced and vanish completely in the "semi-dilute" solution regime (above  $W_2^{2\mu}$ ) for both PI's, with the corresponding activation energies being  $2.1 \cdot 10^3$  for PI-1 and  $1.8 \cdot 10^3$ K for PI-2 (Figure 6). Assuming that viscous flow of aggregates requires higher values of  $\Delta E_v/k$ , it follows from the data obtained that theflow entities in the semidilute solution regime are aggregates of PI chains in a coiled conformation similar to that of individual chains at  $T > 303$  K, rather than aggregates of extended macromolecules.

At still higher polymer concentrations  $(W_2 > W_2^2)$  the non-Newtonian behavior sets in, that behavior being more pronounced at lower temperature (Figure **7).** As can be seen from Figure 8, the double logarithmic plots of shear viscosity (at constant shear rate) against polymer concentration consist of two linear portions intersecting at the third "critical" concentration,  $W_2^{3u}$ , with values of 0.11 for PI-1 and 0.20 for PI-2. A jump-like increase of the slopes on those plots at  $W_2^{3u}$  from 0.5 to 26.0 (PI-1) and from 0.3 to 12.0 (PI-2) is undoubtedly a manifestation of transition to a "concentrated" solution regime, characterized by formation of a continuous network of entangled, coiled macromolecules.

This latter transition to "concentrated" solution regime may be detected not only from log  $\eta$ -versus-log  $W_2$  plots at constant molecular mass,  $M_2$  (Figure 8), but also from log  $\eta$ -versus-log  $M_2$  plots at constant concentration,  $W_2$ . Figure 9 shows that the latter plots also consist of two straight lines intersecting at the "critical" (entanglement) molecular mass,  $M_2^u$ . As expected,  $M_2^u$  is lower the higher is  $W_2$ , these two quantities being connected by an apparent scaling relationship,  $W_2 \propto$  $M_7^{-0.8}$ . This result is in a good qualitative agreement with the corresponding theoretical prediction.<sup>11</sup>

#### **CONCLUSIONS**

**As** follows from the experimental study of thermodynamic properties and flow behavior of water solutions of two model polyionenes, in the "extremely dilute" solution regime below the first "critical" concentration,  $W_2^{\text{1}}$ ", the macromolecules of both PI's assume the most extended conformation due to the effect of intramolecular repulsion between like charges in the chain backbone. This effect gradually diminishes on transition to the "dilute" solution regime  $(W_2^1 \lt W_2 \lt W_2^2)$ since the intramolecular charges of PI's are shielded by counterions. and the chain conformation becomes less extended. Heating the dilute solution above  $T = 303$ K makes the individual PI chains assume a coiled conformation as a result of the structural change in bulk water. Intermolecular shielding of PI charges in the "semidilute" solution regime  $(W_2^2 u < W_2 < W_2^3 u)$  due to a lateral aggregation of macromolecules also promotes their transition into a coiled conformation. Finally, in the "concentrated" solution regime above the third "critical" concentration,  $W_2^{3u}$ , [or, equivalently, above the "critical" molecular mass,  $M_2^u = f(W_2)$ ], the coiled macromolecules form a continuous entanglement network.

It should be remarked here that although the same physical mechanisms were invoked to explain the special features of concentration dependences of both heats of solution and dilution on one hand, and solution viscosity. on the other. the boundaries between different solution regimes deduced from both sets of data did not coincide. Nevertheless. this apparent discrepancy is in fact not too alarming in view of the obviously different sensitivity of both methods to structural changes occurring in the solute and/or in the solvent phase.

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